Supplementary Information

Table S1 Chemical compositions (wt.%) of the 316L SSs investigated in this study. The higher contents of O and N in the SLM alloy primarily originated from the powder sources [S1].

Material	Fe	Cr	Ni	Mo	С	Mn	Р	S	Si	Ο	Ν
SLM	Bal.	16.7	10.8	2.25	0.011	1.20	0.023	0.0049	0.54	0.041	0.170
СМ	Bal.	16.4	10.0	2.05	0.022	1.04	0.030	0.0018	0.40	0.003	0.011

Table S2 Comparison of the H diffusivities in the SLM and CM 316L alloys, which were obtained from gas-phase permeation tests, reported in the literature. The diffusivity of H in the alloy, $D_{\rm H}$, follows the Arrhenius relation, $D_{H} = D_0 \exp(-E_d/RT)$, where D_0 is a material constant, $E_{\rm d}$ is the activation barrier energy for diffusion, R is the universal gas constant (8.314 J·mol⁻¹·K⁻¹), and T is the absolute temperature.

	Manufacturing	D_0	$E_{ m d}$	$D_{\rm H}$ @RT	<i>D</i> н@50 °С	
	method	$[m^2/s]$	[kJ/mol]	$[m^2/s]$	$[m^2/s]$	
Lin <i>et al</i> . [S2]	SLM	1.01×10 ⁻⁶	56.16	1.43×10 ⁻¹⁶	8.28×10 ⁻¹⁶	
Brass & Chêne [S3]	СМ	6.20×10 ⁻⁷	53.63	2.44×10 ⁻¹⁶	1.30×10 ⁻¹⁵	
Lee <i>et al</i> . [S4]		1.24×10 ⁻⁶	55.10	2.69×10 ⁻¹⁶	1.51×10 ⁻¹⁵	
San Marchi et al. [S5]		8.90×10 ⁻⁷	53.90	3.14×10 ⁻¹⁶	1.69×10 ⁻¹⁵	
Kishimoto et al. [S6]		1.30×10 ⁻⁶	54.03	4.35×10 ⁻¹⁶	2.35×10 ⁻¹⁵	



Figure S1 X-ray diffraction (XRD; D/MAX-2500, Rigaku-Denki, Tokyo, Japan) scan results of CM and AM 316L specimen.

References

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